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#### 37 FRONT MATTER

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- 39 **3D photothermal AFM-IR tomography at nanometer scale**
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# 51 Abstract

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Photothermal AFM-IR microscopy enables label-free chemical imaging with nanometer scale 53 spatial resolution through the integration of atomic force microscopy (AFM) and infrared radiation. 54 55 The capability for subsurface and 3D tomographic material analysis remains however largely unexplored. Here, we establish a simple and robust relationship between probing depth and laser 56 repetition rate. Using this relationship, we demonstrate how Photothermal AFM-IR of thin 57 surface/subsurface layers of PS-PMMA blends can result in 3D representations revealing the size 58 and thickness of PS droplets in the PMMA matrix with nanometer scale resolution. Experimental 59 findings are confirmed by analytical models. 60

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# 63 Introduction

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Atomic force microscopy (AFM) is an efficient technique for surface and sub-surface analysis at 65 the nanometer scale, with applications in many scientific domains, including surface science, 66 polymer science, cellular biology, molecular biology, molecular engineering, solid state physics, 67 and medicine. The main advantages of this technique are its simplicity, robustness and the 68 possibility to probe a wide variety of physical properties of the surface directly below the tip of the 69 AFM with nanometer scale spatial resolution. Several AFM techniques have been developed to 70 identify or discriminate the different nature or composition of materials through characterization of 71 the chemical properties at the nanometer scale. During the last decades, two different approaches 72 have been developed to perform unambiguous chemical characterization of samples using infrared 73 spectroscopy: AFM-IR and s-SNOM. These AFM-based techniques allow one to measure the 74 infrared spectra and chemical maps with a few nanometers of lateral resolution (1, 2). Both 75 techniques have advantages and drawbacks and can complement each other dependent on the type 76 77 or nature of the sample to be studied.

Most AFM based techniques are probing the top surface of the sample. A few methods, such as 78 scanning microwave impedance microscopy (SMIM), Kelvin probe force microscopy (KPFM), 79 80 acoustic force microscopy and contact-resonance atomic force microscopy have shown a contrast imaging change related to different probing depths (3–9). The probing depth of SMIM is directly 81 linked to the exponentially decreasing intensity of the microwave electric field from the surface 82 83 into the sample. This effect is called skin effect and determines the probing depth and sampling volume of the measurement at a fixed frequency of analysis. The relationship between microwave 84 frequency and probing depth can lead to sample investigation over various depths and enables one 85 to reconstruct the real shape of buried structures similar to X-ray tomography (10). For acoustic 86

force microscopy the dependency of the probing depth is more complex to determine as the physical 87 process involved is generated by nonlinear interaction between two acoustic waves. Both the 88 acoustic wave frequency and the frequency difference variation result in strong contrast variations 89 pointing to probing depth variations (11). As the AFM-IR technique is based on photothermal 90 excitation, the physics involved should be the same as for conventional photothermal spectroscopy 91 92 and the laser repetition rate used for infrared analysis should control the probing depth of each measurement. 93

94 In this work, we present a full characterization of the probing depth for all commonly used AFM-IR modes: resonance enhanced AFM-IR mode, tapping AFM-IR mode, and surface sensitive AFM-IR 95 mode (12) and propose a methodology to calibrate rigorously this property independently of the 96 type of cantilever used. Thanks to this knowledge, we show, how a 3D tomogram can be 97 reconstructed to visualize the full shape of polymer samples inside a matrix of another polymer, 98 99 based on data acquired in resonance enhanced AFM-IR mode.

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- 102
- **Results** 103

#### 104 Methodology and sample calibration for probing depth estimation 105

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107 The probing depth is mainly defined as a property that represents the depth sensitivity of a surface characterization method. In the case of the AFM-IR technique, it means the ability to probe 108 the sample absorbance below its surface. In classical photothermal techniques, the probing depth is 109 associated with the thermal diffusion length  $L_{diff}$  which is dependent on the frequency (f) of the heat 110 modulation (13): 111

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L <sub>diff</sub>	$\propto \frac{1}{\sqrt{f}}$	(1)
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As the AFM-IR technique is based on photothermal excitation induced by the laser pulses, the 113 114 frequency of the heat modulation is directly given by the laser repetition rate or laser frequency. Recent experiments using resonance enhanced AFM-IR (12, 13) confirmed an increase in probing 115 depth with a decrease in laser repetition rate, although not following Eq (1) quantitatively. 116

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118 Given the current lack of a detailed description of the probing depth, we propose a different approach. In this paper, we first calibrate the probing depth on a model sample consisting of two 119 120 different polymers: polymethylmethacrylate (PMMA) and polystyrene (PS), forming a PS-PMMA calibration wedge. This calibration then serves as the input for a tomographic reconstruction of the 121 122 sample of interest: a PS-PMMA blend (see Materials and Methods).

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The PS-PMMA calibration wedge sample is made of a homogeneous PMMA film with a thickness of about 18 µm onto which a 2 µm thick PS droplet of 200 µm diameter has been deposited 125 (Fig. 1A). The edge of the PS droplet forms a wedge of PS on the PMMA substrate and simulates 126 a well-defined, monotonic increase in PS sample thickness in the AFM-IR analysis. For the 127 calibration, the AFM cantilever is scanned perpendicularly to the PS wedge to avoid coupling of 128 lateral or torsional forces into the normal (vertical) force detection during the contact mode AFM 129 scan (Fig. 1B). 130



Fig. 1 Calibration sample and setup A) Schematics of the tip and the PS wedge on the thick PMMA film.
B) Optical microscope image of the PS-PMMA calibration wedge sample. A sketch of the Au-coated cantilever with the tip on the bottom end is shown and the arrows indicate the scan direction. C) Contact resonance modes of the HQ cantilever in contact mode obtained on the PMMA film with the laser tuned to the C=O absorption band at 1730 cm<sup>-1</sup>. The modes are: 68 kHz, 213 kHz, 426 kHz, 644 kHz, 869 kHz, 1236 kHz, 1709 kHz, 2269 kHz, 2912 kHz.

141 We describe in the following the depth calibration process. Resonance enhanced AFM-IR is the employed technique that is later compared to surface sensitive and tapping AFM-IR. In our 142 resonance enhanced AFM-IR measurements, we selected a HQ:CSC38/Cr-Au (Mikromasch) 143 contact mode cantilever with 0.03 N/m spring constant and Cr-Au coating on both sides that can be 144 145 excited at 9 different contact resonance modes over the 0-3 MHz range (Fig. 1C). For all the experiments, the calibration process is realized on the exact same location of the PS wedge to avoid 146 147 any variation of the absorption signal from topography (different slope) or change of mechanical interaction between the tip and the sample. A phase locked loop (PLL) is activated to track any 148 change of the contact resonance during the AFM-IR scan, thus automatically adjusting the laser 149 repetition rate to always match the local contact resonance. Most of the time, the phase shift 150 corresponds to a cantilever resonance shift related to the mechanical contact change between the 151 PMMA and the PS surface. The use of a frequency-tracking mechanism is crucial in order to obtain 152 artifact-free AFM-IR signals that could otherwise overestimate the absorbance of either the PMMA 153 or PS domain. Furthermore, the choice of materials for wedge and substrate is critical for the most 154 accurate calibration of the probing depth. For example, using a substrate of silicon together with a 155 PMMA wedge leads to strong phase shifts that might challenge the PLL to correct properly, 156 resulting in an overestimated probing depth (Fig. S1). This large phase shift is directly related to 157 the fact that the Young's modulus of silicon is ~30 times greater than that of PS. This illustrates the 158 importance of a perfectly suited substrate to reduce the mechanical variation between the wedge 159 and substrate material. In all our experiments on the PS-PMMA calibration wedge sample, the PLL 160 easily corrects for the phase shift as their Young's moduli are close and equal to 3.1 GPa and 2.7 161 GPa for PMMA substrate and PS wedge, respectively. Only on the first nanometers of the tip of the 162

PS wedge, the phase shift is not properly compensated by the PLL, probably because of a change
of lateral friction when passing from PMMA to PS (see Fig. S2). This very small fraction of the PS
wedge is systematically excluded in the probing depth analysis.

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In the calibration process, we first scan for each cantilever mode (see Fig. 1C), i.e. laser repetition 167 rate, the topography together with the AFM-IR signal at 1730 cm<sup>-1</sup> which is the main absorption 168 peak of the PMMA carbonyl band. Following the PMMA substrate absorption signal for a specific 169 170 cantilever mode along the wedge allows to evaluate at which depth, or PS thickness, the buried PMMA can be detected by the AFM-IR technique, and then to estimate the probing depth of the 171 technique. Any tilt in the topography image of the exposed PMMA substrate area is corrected with 172 a plane fit. From the averaged line profiles across the wedge (10 lines were averaged, see Fig. 2A 173 and Fig. 2D for a laser repetition rate of 610 kHz and 1235 kHz, respectively) the height offset is 174 removed to arrive at the relation between the x axis position and the PS thickness. With the real PS 175 176 thickness as a function of the x-position, it is then possible to represent the simultaneously recorded AFM-IR signal as a function of the PS thickness. Taking into account the imperfect correction of 177 the PLL due to friction as mentioned above, the first nanometers of (PS) thickness have been 178 179 removed from the graphs. As shown in the resulting Fig. 2C and Fig. 2F, representing respectively the thickness dependent AFM-IR signals at 610 kHz and 1235 kHz, the behavior of the AFM-IR 180 signal is very well fitted by an exponential decrease law demonstrating that the thermal expansion 181 shows similarities to the exponential nature of the simplest form of Beer's law used in computed 182 tomography, as well as the penetration depth of microwave power. For all cantilever modes the 183 mathematical fit formula can be written as: 184

$$A(z) = m_1 + m_2 e^{-\frac{z}{m_3}}$$
(2)

186 Where z is the wedge thickness below the tip position, and  $m_1$ - $m_3$  are fit parameters.

As the stress created by the thermal expansion decreases exponentially with the distance (or thickness), the probing depth may be defined as the exponential argument ( $m_3$ ), at which the AFM-IR signal decays to 1/e above an offset  $m_1$ . For these two example frequencies in Fig. 2C and Fig. 2F, we can confirm that the probing depth decreases as the mode frequency increases: 292 nm for 610 kHz and 125 nm for 1235 kHz



**Fig.2 Methodology to estimate the probing depth of PMMA at 610 kHz and 1235 kHz.** A) Topography line profile of the PS wedge. B) Corresponding AFM-IR signal at 610 kHz for 1730 cm<sup>-1</sup>. C) AFM-IR signal at 610 kHz as a function of PS thickness (red curve), the dashed blue curve represents the corresponding exponential fit. The table gives the fit values. D) Topography line profile of the PS wedge. E) Corresponding AFM-IR signal at 1235 kHz for 1730 cm<sup>-1</sup>. F) AFM-IR signal at 1235 kHz as a function of PS thickness (red curve), the dashed blue curve represents the corresponding exponential fit. The table gives the fit values.

The same study is repeated using the PS absorption band at 1600 cm<sup>-1</sup> to evaluate the probing depth 200 for top-layer absorption (Fig. 3) compared to the previous buried substrate absorption (PMMA, 201 202 Fig. 2). The difference of this new case is that the tip is now directly on the expanding sample when on the PS. Opposite to probing the buried PMMA substrate before, we now observe an AFM-IR 203 signal increase with increasing PS wedge thickness. The aromatic C=C absorption band of PS is 204 much weaker than the carbonyl C=O absorption band of PMMA. To compensate for this difference 205 in absorption strength, the laser power has been increased to 4.9% (instead of 0.5% for PMMA). 206 The data treatment follows the exact same procedure as before (tilt removal and offset correction 207 208 of the topography) but the fit function used to describe the AFM-IR signal of the PS droplet is now given by the following expression: 209

$$A(z) = m_1 + m_2 \left( 1 - e^{-\frac{z}{m_3}} \right)$$
(3)

- 210 211
- 212

213 Where z is the wedge thickness at the tip position, and  $m_1$ - $m_3$  are fit parameters.

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**Fig. 3** *Methodology to estimate the probing depth of PS.* A) Topography line profile of the PS wedge. B) Corresponding AFM-IR signal at 2272 kHz for 1600 cm<sup>-1</sup>. C) AFM-IR signal at 2272 kHz as a function of PS thickness (red curve) and exponential fit (dashed blue curve). Fit parameters are given in the table. D) Topography line profile of the PS wedge. E) Corresponding AFM-IR signal at 2915 kHz for 1600 cm<sup>-1</sup>. F) AFM-IR signal at 2915 kHz as a function of PS thickness (red curve) and exponential fit (dashed blue curve). Fit parameters are given in the table.

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# 227 Probing depth calibration in resonance enhanced contact mode AFM-IR

Above calibration procedure is carried out for all 9 accessible contact resonances (Fig. 1C) of the Mikromasch HQ probe in resonance enhanced AFM-IR. The probing depth, i.e. fit parameter  $m_3$ in Eq (2) for PMMA and (3) for PS, respectively, is reported in Fig. 4 as a function of the laser frequency.



Fig. 4 Probing depth as a function of the laser frequency. PMMA is represented by red dots and PS by blue squares, error bars are represented in the corresponding color, the dashed curves indicate the corresponding fits and the orange curve corresponds to the thermal diffusion length.

The obtained curves for PMMA and PS are compared to the thermal diffusion length defined by the power law of Eq. (1) with a power-law exponent of -1/2. The comparison clearly shows that the decrease of experimental probing depth as a function of the laser frequency is much faster than for a power law exponent of -1/2 for the thermal diffusion length. The best fit function found to describe this behavior can be expressed by the formula below and table 1 reports the fitted values for PMMA and PS with both a correlation coefficient R=0.998:

246

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$$PDepth(f) = \frac{a}{(f-b)^c}$$
(4)

247 Where f is the laser frequency

	PMMA	PS
а	2.4 1011	2.9 1011
b (Hz)	-343 000	-200 000
с	1.499	1.55

Table 1 : Fitted values for PMMA and PS.

251 The power argument (c) extracted from the fit function (4) for both cases seems to be close to -3/2. This could be explained by the fact that the total transfer function is simply the product between the thermal 252 diffusion (power law exponent -1/2) and the response of the cantilever (exponent -1). The extrapolation of 253 254 these laws to a laser frequency of 1 kHz allows us to predict the probing depth of the first AFM-IR implementation (not resonantly-enhanced) using a low-repetition rate OPO laser. The probing depth are 255 256 respectively 1.2 µm for PMMA and 1.56 µm for PS. These values are in good agreement with the 1 µm range of linearity experimentally estimated by Centrone and coworker using a PMMA wedge on zinc 257 selenium (14). Indeed, the range of linearity of the contact mode must be much smaller than the probing 258 259 depth and correspond to the range where the exponential decay can be fitted by a linear dependency.

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# 262 **Probing depth calibration in surface sensitive AFM-IR and tapping AFM-IR mode**

Surface sensitive AFM-IR is a relatively new operating mode that allows one to reduce drastically 264 the probing depth of contact mode (12). The calibration in this previous work has been realized on 265 a sample (polyurethane on silicon substrate) not specifically designed to avoid mechanical artifacts. 266 The methodology described in the previous paragraph is used to reevaluate the probing depth on 267 the optimized PS-PMMA calibration wedge. The mechanical oscillation of the tip is fixed at 3 MHz 268 and generated by the piezo actuator (the one typically used in AFM tapping mode). The laser 269 frequency is tuned to match the difference of piezo and laser frequencies with a contact resonance 270 mode of the cantilever. This frequency mixing approach is similar to DFG (difference frequency 271 generation) and possible only because the tip-surface interaction is non-linear (12). 272

273 The probing depth of the surface sensitive mode is represented in Fig.5.



# Fig. 5 Surface sensitive probing depth as a function of the laser frequency. Red and blue data points represent respectively the results for PMMA and PS for surface sensitive AFM-IR. Error bars are

277 represented in corresponding colors. Orange and green squares represent the probing depth of the tapping
278 AFM-IR mode for PMMA and PS, respectively.
279

The probing depth of the surface sensitive mode seems to be equivalent for both PS and PMMA polymers at high laser frequencies. Note that in our experiment the PLL is activated and compensates perfectly for mechanical effects at the interface of the two polymers.

Furthermore, during the experiment, it was also quite difficult to go below a 1 MHz laser frequency because of instabilities in the IR signal. But even in this limited laser frequency range, we can see that the probing depth decreases from 50 nm to 10 nm. The quality of the wedge is still a limitation to the accuracy of this experiment. The first nanometers of the wedge are important for a good estimation of the probing depth as we need nanometer sensitivity. This is why an improved design of the wedge more adapted for small thicknesses should give us a more accurate view of the evolution of the probing depth in the range of 2 MHz -3 MHz.

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291 Tapping AFM-IR mode is also a surface sensitive mode. The only difference is that instead of modulating the tip motion continuously on the surface in constant contact with the sample (contact 292 resonance), the tip is now in intermittent contact in tapping AFM-IR. Based on the same non-linear 293 interaction, the theory used to describe the measurement is the same for both cases (12, 15, 16). 294 Using the same PS-PMMA calibration sample as for the surface sensitive and resonance enhanced 295 AFM-IR; the probing depth has been evaluated for a 300 kHz tapping gold coated cantilever 296 PPP-NCHAu-MB-10 from Nanosensors (Fig.5). The cantilever mode used for the piezo drive was 297 fixed at 1694 kHz and the mode to detect the IR signal was fixed at 272 kHz. In order to minimize 298 mechanical induced artifacts during scanning, two PLLs were used to track both the tapping 299 frequency and the IR detection frequency. The probing depth evaluated under these conditions is 300 about 30 nm (Fig.S3) which is considerably smaller than our previous evaluation of 50 nm for the 301 polyurethane wedge on Si substrate (12). This confirms the necessity to apply the same and correct 302 methodology to compare the probing depth properties of different polymer nature. At the same laser 303 frequency, the probing depth for tapping AFM-IR matches well with the probing depth for surface 304 sensitive AFM-IR attesting that the physics involved in the non-linear interaction is the same. 305

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Compared to the data presented in Fig. 4 for resonance enhanced AFM-IR, surface sensitive AFM-IR and tapping AFM-IR provide a smaller probing depth (i.e. higher surface sensitivity) at equivalent laser frequencies. We attribute this to the non-linear interaction for the latter two modes.

# 311 **Tomography reconstruction**

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After above depth calibration, we now turn to its application of subsurface sensing in tomography. 313 The sample used to illustrate the potential of tomography reconstruction is a blend of PS-PMMA 314 presented in the material and method section. This sample is a matrix of PMMA of 1.9 µm thickness 315 wherein PS droplets of different sizes are embedded. The mode of operation is resonance enhanced 316 AFM-IR. Using the exact same HQ:CSC38/Cr-Au cantilever as for the calibration, we have 317 recorded the chemical maps at 1730 cm<sup>-1</sup> corresponding to the ester carbonyl absorption band of 318 PMMA for all the contact resonance modes from 50 kHz to 3 MHz (9 modes, Fig. 1C). Figure 6 319 320 represents these chemical maps of the same area where the darker PS domains are surrounded by a bright, absorbing PMMA matrix (corresponding topography in Fig. S4). 321



FIG. 6 : Chemical maps of the PS-PMMA blend at the ester carbonyl absorption band of PMMA
(1730 cm<sup>-1</sup>) for all the contact resonance modes. A) Laser frequency 68 kHz. B) 212 kHz. C) 417 kHz. D)
620 kHz. E) 870 kHz. F) 1235 kHz. G) 1705 kHz. H) 2275 kHz. I) 2911 kHz. The white scale bar corresponds
to 750 nm.

The mixing of PS and PMMA resembles the one of oil and water, respectively, where PS similar 328 329 to oil minimizes its thickness while floating on PMMA. The full thickness of the PMMA matrix is 1.9 µm which is larger than the expected thickness of the PS droplet. Consequently, there should 330 be a non-negligible amount of PMMA below all the PS droplets. In the measurements presented in 331 332 Fig. 6, we clearly see that when the laser frequency increases, the signal contrast between the PS droplets compared to the PMMA matrix also increases, i.e. while PS and PMMA areas show a 333 nearly comparable signal strength at 68 kHz, the PMMA signal at 2911 kHz in PS droplet is greatly 334 reduced. The increase of PS contrast is directly related to the probing depth: as the laser frequency 335 increases, the amount of PMMA seen below the PS decreases and the resulting contrast increases. 336 Note that, based on the obtained PS contrast that appears relatively uniform within each droplet, 337 the droplets seem to be not spherical but rather shaped more like puddles. We can also detect very 338 small nanoparticles that are not absorbing at 1730 cm<sup>-1</sup> with a size about 50 nm diameter (Fig. S4 339 and S5). These nanoparticles are neither PS nor PMMA but some contamination of the PMMA 340 solvent and have a strong carbonyl absorption band around 1650 cm<sup>-1</sup> (Fig. S5). 341 342

To reconstruct a 3D tomogram of our sample, it is necessary to express the AFM-IR intensity 343 analogue to Eq. (2) and include the thickness of PS as a function of the position x, y on the surface: 344

346 347

$$I(x, y, f_n) = I_0(f_n)e^{-z(x, y)/p_n}$$
(5)

Here,  $I_0(f_n)$  corresponds to the AFM-IR signal on pure PMMA (z=0) for a cantilever contact 348 resonance frequency  $f_n$ , z(x,y) is the thickness of PS at the position x,y (z=0 on pure PMMA), and 349  $p_n$  is the probing depth, evaluated in the previous paragraph, at frequency  $f_n$ . This expression is 350 valid only for the case of PMMA absorption and PS on the top which correspond to the IR maps 351 resented Fig.6. 352

354 In order to reconstruct the tomogram, i.e. the three-dimensional sub-surface distribution of the PS-PMMA blend, we can use the inverse transformation of Eq. (5) for each pixel of the AFM-IR maps, 355 and for each resonance of the cantilever: 356

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$$z(x,y) = -p_n ln\left(\frac{l(x,y,f_n)}{I_0(f_n)}\right) \quad (6)$$

We assume that there is no PMMA layer on the PS samples. This is confirmed by the surface 360 sensitive maps (Fig. S6.) that give zero level of absorption at 1730 cm<sup>-1</sup>, meaning that the top 361 surface of PS sample is only PS. Applying this inverse transformation on the cross section through 362 the large PS droplet (white dashed line in Fig. 6I) for all the contact resonance modes we obtain the 363 reconstruction of the PS droplet shape in depth or thickness z (Fig. 7). Our previous calibration data 364 obtained at 1730 cm<sup>-1</sup> (PMMA absorbs) in Fig. 4 entered as the probing depth p<sub>n</sub> (see the 365 corresponding values in the legend of Fig. 7). 366

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- 374 With this reconstruction we clearly demonstrate that the 4 first modes (68 kHz 620 kHz) are not
- suited for the tomography reconstruction of this sample because they do not converge to the same
- 376 value as for the higher modes. Since the thickness of PMMA is finite (1.9  $\mu$ m) these 4 lowfrequency modes probe below the PMMA layer and interact with the underlaying silicon substrate.
- This sampled silicon volume is not absorbing at  $1730 \text{ cm}^{-1}$ , therefore, does not contribute to an
- 379 AFM-IR signal in the same way as a thicker PS droplet would. Hence, these 4 lower modes
- 380 overestimate the PS thickness. Moreover, the lower modes have lower lateral resolution and
- therefore also interact with the surrounding PMMA. For frequencies equal and higher than 870
- 382 kHz, the calculated PS droplet thickness seems to converge around a thickness of 138 nm for the
- 383 line profiles of Fig. 7. Consequently, the reconstruction can be applied to the entire scan of the
- surface by considering only these higher modes from 870 kHz to 2911 kHz and averaging the 5
   corresponding images. This gives a good approximation of the size of the PS droplet with a small
   error induced by this type of reconstruction.
- This averaged reconstruction of the 1730 cm<sup>-1</sup> maps revealing the three-dimensional shapes of the PS droplets is represented in Fig. 8.



Fig. 8: 3D representation of the PS-PMMA blend. A) 3D representation revealing the size & thickness of
 PS droplets . B) Same data as in A) but with inverted z-axis and rotated for clarity. Some PS droplets are
 labelled P1-P5.

We can clearly distinguish several droplets that are in fact shaped more like puddles than droplets as the thickness is around one hundred nanometers or less at a size of a few micrometers. The accuracy of the calculated thickness is estimated by the analysis of all the modes and reveals a quite good standard deviation of a few nanometers. Table 2 summarizes the maximum thickness of a few puddles (P1-P5) indicated in Fig. 8B.



Averaged	138	53	58	37	22
thickness (nm)					
Error (nm)	6	5	6	6	3

399 Table 2 : Thickness of representative PS droplets from Fig. 8B.

We can also observe in Fig. 8 that the nanoparticle surface contaminants have a characteristic thickness of about 70 nm for a diameter of around 60 nm and a height of 10 nm. The thickness value is probably overestimated because the probing depth calibration has been realized with PS and not with the contaminant. We can observe that the constraint of the outlined 3D reconstruction procedure is limited to give reliable depth information only for those materials that were used for the probing depth calibration, i.e. here, the PMMA matrix and the embedded PS droplets.

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So far, we utilized the depth calibration based on the 1730 cm<sup>-1</sup> PMMA absorption, see Fig. 4. Carrying out the 3D tomography reconstruction of PS based on the depth calibration data obtained at the 1600 cm<sup>-1</sup> PS absorption (Fig. 4) for an image of the PS-PMMA blend taken at 1600 cm<sup>-1</sup> is more complicated. Analogue to Eq. (6) we can derive an expression for the position-dependent PS thickness z:

 $z(x, y) = -p_n ln \left( 1 - \frac{l(x, y, f_n)}{l_{0_P S}(f_n)} \right)$ (7)

Now,  $I_{0 PS}(f_n)$  corresponds to the AFM-IR signal on pure PS (z=0). While before, the AFM-IR 417 signal on pure PMMA,  $I_0(f_n)$ , was accessible in the 1730 cm-1 scans (Fig. 6) due to the thick PMMA 418 matrix,  $I_{0 PS}(f_n)$  cannot be estimated from the 1600 cm-1 scans since its value corresponds to an 419 infinite PS thickness, or a PS thickness much larger than the probing depth. But the thickest PS 420 droplet is only ~138 nm thin and hence there is no way to estimate the value  $I_{0_{PS}}(f_n)$  for the given 421 cantilever modes. I<sub>0 PS</sub>( $f_n$ ) can be an adjustable parameter to converge to the estimated thickness of 422 the P1 droplet, the 138 nm reconstructed from the PMMA absorption scans. The 3D reconstruction 423 with PS absorption maps is represented in SI (Fig S7) but does not give any additional information 424 compared to the previous 3D tomography for the PMMA absorption. 425

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#### 427 428

# 429 Discussion

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In this work we have described a robust methodology to calibrate the probing depth of AFM-IR imaging modes (contact, surface sensitive and tapping). With this calibration it was then possible to reconstruct in 3D the tomography of a polymer blend and hence to determine the sub-surface 3D shape of polystyrene puddles in a polymethacrylate matrix.

The calibration presented here has only been realized for a single cantilever but it would be 435 interesting in a next step to repeat with cantilevers from the same batch to learn about their statistical 436 dispersion and see the possibility to establish a more general calibration curve. Other AFM-IR 437 cantilever types and brands can also be characterized regarding their probing depth. This type of 438 study would be long and fastidious, but it would surely be very useful for the community to obtain 439 such reference values for the probing depth for a known polymer. Our work here only concerns PS 440 and PMMA but the method is applicable to other polymers as well, as they can have orthogonal 441 solvents for the drop cast preparation, or one can imagine other ways to create a clean calibration 442 wedge. We have seen that the wedge obtained with the drop cast method is not optimal for surface 443 444 sensitive AFM-IR as we need a smaller slope of the wedge to get a better estimation for the smaller probing depths (10-20 nm). However, the drop cast wedge is still a good standard to evaluate theprobing depth of contact mode of any kind of polymer material.

We have also pointed out that it is crucial to minimize any mechanical impedance between the calibration wedge and its substrate to obtain a good estimation of the probing depth. However, the calibration wedge must also be matched to the material system for which the 3D tomography reconstruction is carried out. For example, the calibration data for a PMMA wedge on a Si substrate proved inadequate for our PS-PMMA blend but could be useful for other situations where the sample is deposited on Si substrate. This necessity to design a suitable wedge for each sample of interest is a limitation of the presented calibration procedure.

The calibration methodology developed in this study (wedge-substrate analysis) allows an 454 estimation of the probing depth for an expanding sample, buried or not. For this, we assume that 455 the signal at the wedge position (i.e. a given thickness of PS) is perfectly equivalent for a flat layer 456 of the same thickness in bulk. The probing depth was defined as an AFM-IR signal drop to 1/e 457 (~37%), i.e. it captures 63% of the AFM-IR signal, but it does not give the range in which the 458 technique is able to "feel" the sample. A sampling depth could be defined as 3 times the probing 459 depth to comprise 95% of the AFM-IR signal, which would be an alternative representation of the 460 depth sensitivity of the respective cantilever mode. Turning to Fig. 4, the empirical law from our 461 measurement fits very well the behavior of the cantilever modes and the sampling depth can be 462 easily evaluated as a function of the mode frequency: at 3 MHz the sampling depth is around 463 150 nm and for 200 kHz it is 3 µm. This law should be only dependent on the laser frequency, and 464 neither on the power nor the pulse duration. Further work is required to fully map out th impact of 465 all the laser parameters on the probing depth. 466

The empirical law is a power law with an exponent of -3/2 which is different from -1/2 for the classical thermal law. This is probably related to the fact that we are measuring a photothermal expansion and this phenomenon is not only thermal but also mechanical, i.e. the transfer function between tip and cantilever has also a strong influence.

The outlined probing depth calibration procedure allowed us to introduce a novel application of the 471 AFM-IR technique: 3D tomography with sub-surface imaging. We have shown how to reconstruct 472 the three-dimensional shape of PS puddles in a PMMA matrix of a PS-PMMA blend with very 473 good accuracy (only a few nanometers of error). This 3D reconstruction is only possible using a 474 calibration sample with a PMMA matrix covered by a PS wedge. Furthermore, successful 3D 475 tomography requires to calibrate the probing depth based on the matrix (i.e. PMMA) AFM-IR 476 absorption signal: The lack of absorption reveals the real shape of the embedded material (PS) 477 inside the matrix. Calibration based on the PS wedge absorption proved less successful to 478 reconstruct the 3D shape of the sample of interest since its surface contains no PS puddles of infinite 479 or very large thickness (with respect to the probing depth) that are required by the reconstruction 480 formula. As another drawback, this 3D tomography is currently not applicable to all samples and 481 will be restricted to binary blends. We have not yet devised an experiment to reconstruct a ternary 482 blend, for example 2 different embedded polymers inside another matrix polymer. 483 484

Irrespective of the mentioned potential limitations, the 3D tomography is a property only reserved 485 for the AFM-IR technique as it is a photothermal measurement in which the perfect control of the 486 probing depth allows the depth investigation. This approach is not possible in sSNOM even if many 487 works have shown 3D maps of electromagnetic fields of the tip-sample interaction (17). The 488 scattering of the light by the tip is imposed by its shape and there is currently no practical way to 489 change the evanescent wave generated by the tip to modulate the electric field probing depth which 490 491 is around 10-50 nm (18). The biggest advantage of AFM-IR is its flexibility to probe in a depth ranging from several micrometers to a few nanometers just by changing the laser frequency from 492 493 tens of kHz to a few MHz.

#### 496 Materials and Methods

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#### 498 Sample preparation

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500 Polymethylmethacrylate (PMMA) (150 kDa) and polystyrene (PS) (50 kDa) were purchased from 501 Sigma-Aldrich. Those polymers were chosen for their comparable mechanical rigidity properties

- Solution of the s
- Various solutions were prepared: a blend of PS and PMMA in THF with a concentration of 10 mg/mL, a solution of PMMA in THF with a concentration of 10 mg/mL, and a solution of PS in cyclohexane with a concentration of 1 mg/mL.
- 507 Sample preparation involved the solvent casting method. The PS-PMMA blend and the PMMA 508 solutions were dropped on a silicon wafer under a controlled atmosphere. Subsequently, for the PS
- wedge on PMMA samples, a few droplets of PS solution were deposited over the PMMA thin film.
   The samples were then annealed at 125°C for 6 hours after each step of the deposition to remove
- 510 The samples were then annealed at 125°C for 6 hours after each step of the deposition to remove 511 all traces of solvent. By using this simple method, the polymer layer thicknesses are about a few 512 micrometers.
- 512 micror 513

# 514 **AFM**

The AFM-IR system is a Dimension IconIR from Bruker with an infrared tunable QCL laser from Daylight Solutions covering the 1900-900 cm<sup>-1</sup> range. The pulse length is fixed at 100 ns for all the experiments. Power is fixed at 0.5% for the PMMA analysis at 1730 cm<sup>-1</sup>, and at 4.9% for PS

at  $1600 \text{ cm}^{-1}$ . The cantilever used in resonance enhanced AFM-IR (contact mode) and surface

sensitive AFM-IR is a cantilever from Mikromasch: HQ:CSC38/Cr-Au with CR/Au coating on

- 520 both sides and a spring constant of 0.03 N/m. The cantilever used for tapping AFM-IR is from
- 521 Nanosensors reference PPP-NCHAu-MB-10. Method is also applicable to alternative probe 522 models.
- 522 523

# 524 Imaging analysis

Imaging analysis and treatment has been done by MountainsMap software (version 9) fromDigital Surf.

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# 529 **References**

- 530
- A. Centrone, Infrared Imaging and Spectroscopy Beyond the Diffraction Limit. *Annual Review of Analytical Chemistry* 8, 101–126 (2015).
- A. Dazzi, C. B. Prater, AFM-IR: Technology and Applications in Nanoscale Infrared Spectroscopy and Chemical Imaging. *Chem Rev* 117, 5146–5173 (2017).
- 5353.M. Tabib-Azar, Y. Wang, Design and fabrication of scanning near-field microwave probes compatible with536atomic force microscopy to image embedded nanostructures. *IEEE Trans Microw Theory Tech* **52**, 971–979537(2004).
- K. Lai, M. B. Ji, N. Leindecker, M. A. Kelly, Z. X. Shen, Atomic-force-microscope-compatible near-field
   scanning microwave microscope with separated excitation and sensing probes. *Review of Scientific Instruments* 78 (2007).
- 5. E. Chilla, T. Hesjedal, H.-J. Fröhlich, Nanoscale determination of phase velocity by scanning acoustic force microscopy. *Phys Rev B* 55, 15852–15855 (1997).
- 5436.U. Rabe, W. Arnold, Acoustic microscopy by atomic force microscopy. Appl Phys Lett 64, 1493–1495544(1994).
- 545 7. K. Yamanaka, H. Ogiso, O. Kolosov, Ultrasonic force microscopy for nanometer resolution subsurface
  546 imaging. *Appl Phys Lett* 64, 178–180 (1994).

547 548	8.	G. Stan, C. V. Ciobanu, S. W. King, Resolving the Subsurface Structure and Elastic Modulus of Layered Films via Contact Resonance Atomic Force Microscopy. <i>ACS Appl Mater Interfaces</i> <b>14</b> , 55238–55248
549 550	0	(2022). K. Koja, A. Assoum, P. Da Wolf, F. Biguamal, A. Nahmaa, A. Naja, T. Bayrouthy, M. Joujad, 2D Imaging
551	9.	K. Kaja, A. Assouili, F. De Woll, F. Fiquelliai, A. Nellinee, A. Naja, T. Deylouiliy, M. Joulau, 5D illiagilig
552		Microscony Adv Mater Interfaces doi: 10.1002/admi.202300503 (2023)
553	10	C Plassard E Bourillot I Rossignol Y Lacroute E Lepleux L Pacheco E Lesniewska Detection of
554	10.	defects buried in metallic samples by scanning microwave microscopy. <i>Phys Rev B</i> <b>83</b> , 121409 (2011).
555	11.	P. Vitry, E. Bourillot, C. Plassard, Y. Lacroute, E. Calkins, L. Tetard, E. Lesniewska, Mode-synthesizing
556		atomic force microscopy for 3D reconstruction of embedded low-density dielectric nanostructures. <i>Nano Res</i>
557		<b>8</b> , 2199–2205 (2015).
558	12.	J. Mathurin, A. Deniset-Besseau, D. Bazin, E. Dartois, M. Wagner, A. Dazzi, Photothermal AFM-IR
559		spectroscopy and imaging: Status, challenges, and trends. J Appl Phys 131 (2022).
560	13.	L. Quaroni, Understanding and Controlling Spatial Resolution, Sensitivity, and Surface Selectivity in
561		Resonant-Mode Photothermal-Induced Resonance Spectroscopy. Anal Chem 92, 3544–3554 (2020).
562	14.	G. Ramer, V. A. Aksyuk, A. Centrone, Quantitative Chemical Analysis at the Nanoscale Using the
563		Photothermal Induced Resonance Technique. Anal Chem 89, 13524–13531 (2017).
564	15.	J. Mathurin, E. Pancani, A. Deniset-Besseau, K. Kjoller, C. B. Prater, R. Gref, A. Dazzi, How to unravel the
565		chemical structure and component localization of individual drug-loaded polymeric nanoparticles by using
566		tapping AFM-IR. Analyst 143, 5940–5949 (2018).
567	16.	J. Mathurin, A. Deniset-Besseau, A. Dazzi, "Advanced infrared nanospectroscopy using photothermal
568		induced resonance technique, AFMIR: New approach using tapping mode" in Acta Physica Polonica A
569		(Polish Academy of Sciences, 2020)vol. 137, pp. 29–32.
570	17.	H. Wang, L. Wang, D. S. Jakob, X. G. Xu, Tomographic and multimodal scattering-type scanning near-field
571		optical microscopy with peak force tapping mode. Nat Commun 9 (2018).
572	18.	T. Taubner, F. Keilmann, R. Hillenbrand, Nanoscale-resolved subsurface imaging by scattering-type near-
573		field optical microscopy. <i>Opt Express</i> 13, 8893 (2005).
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596		spectroscopy Alexandre Dazzi is a conventor of AFM-IR patents licensed to Bruker
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397 500		mano Surraces Division. The other authors state no conflict of interest.
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- Data and materials availability: All data, code, and materials used in the analyses must be 600 available in some form to any researcher for purposes of reproducing or extending the 601 analyses. Include a note explaining any restrictions on materials, such as materials transfer 602 agreements (MTAs). Include accession numbers to any data relevant to the paper and 603 deposited in a public database; include a brief description of the dataset or model with the 604 number. The DMA statement should include the following: "All data are available in the 605 main text or the supplementary materials." 606 607 608 609 **Supplementary Materials** 610 611 Please use the Science Advances template to format your Supplementary Materials. 612
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